

helix axes, thus contributing a rotation in the opposite sense, *i.e.* laevorotatory. The  $RS1/4$  sulphate helix also contributes optical rotation in a laevorotatory sense, but because of the larger sulphate-sulphate distances, to a much lesser extent. The counteracting effect of the  $NH_4$  helix is also small because of the greater  $NH_4$  to  $NH_4$  distances. Models 1 and 2 of Table 4 show that on reducing the polarizability of  $NH_4^+$ , the optical rotation is increased, thus showing how they act in an opposite sense with respect to the other helices as far as the optics is concerned. Thus, despite the complexity of this structure, the simple theory seems to work well for the (100) section, which is also the section with the smallest linear birefringence.

It is far more difficult to apply the simple theory to the (010) and (001) sections, as these have large linear birefringences, which dominate over the optical rotation. While it is possible to find some evidence of helical arrangements of polarizable atoms in these sections, it is, therefore, far from clear how they relate to the sense of optical rotation.

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## Structure of the Hofmann Clathrates $Ni(NH_3)_2Ni(CN)_4 \cdot 2C_6D_6$ and $Zn(NH_3)_2Ni(CN)_4 \cdot 2C_6H_6$

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### Abstract

Rietveld analyses of the neutron powder diffraction patterns of  $Ni(NH_3)_2Ni(CN)_4 \cdot 2C_6D_6$  and  $Zn(NH_3)_2Ni(CN)_4 \cdot 2C_6H_6$  have been performed. The space

group  $P4/m$  is confirmed for  $Ni(NH_3)_2Ni(CN)_4 \cdot 2C_6D_6$  with  $a = 7.2196$ ,  $c = 8.1007$  Å at 25 K,  $a = 7.2358$ ,  $c = 8.3104$  Å at 300 K. The same space group is found for  $Zn(NH_3)_2Ni(CN)_4 \cdot 2C_6H_6$  with  $a = 7.3294$ ,  $c = 8.0722$  Å at 25 K. The effects of

quantum free-rotation of the  $\text{NH}_3$  groups is seen clearly in the temperature factors of the ammine protons.

### 1. Introduction

The Hofmann clathrates (Hofmann & Hocchlen, 1903) have the general formula  $M(\text{NH}_3)_2M'(\text{CN})_4 \cdot 2G$ , where  $M$  is a six-coordinate and  $M'$  a four-coordinate divalent metal ion, and  $G$  is an aromatic guest molecule. There has been renewed interest in these compounds since it has been reported that in  $\text{Ni}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2\text{C}_6\text{D}_6$  the  $\text{NH}_3$  groups undergo near quantum free-rotation at 1.8 K (Wegener, Bostoen & Coddens, 1990) and several other Hofmann clathrates have since been found to show the same phenomenon. Quantum free-rotors in molecular crystals are extremely rare and it is important that the crystal structure of these salts at low temperature be known as completely as possible to enable reasonable dynamical models to be developed.

Previous structure determinations at room temperature were performed more than 20 years ago (Rayner & Powell, 1952; Sasaki, 1969), using X-ray diffraction, but H-atom positions were not determined. In the present paper we refine the crystal structures of  $\text{Ni}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2\text{C}_6\text{D}_6$  at 25 and 300 K and  $\text{Zn}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2\text{C}_6\text{H}_6$  at 25 K. We will refer to these salts as Ni-Ni. $\text{C}_6\text{D}_6$  and Zn-Ni. $\text{C}_6\text{H}_6$ , respectively.

### 2. Experimental

Samples of Ni-Ni. $\text{C}_6\text{D}_6$  and Zn-Ni. $\text{C}_6\text{H}_6$  were prepared and purified according to the procedure given by Baur & Schwarzenbach (1960). The polycrystalline samples were placed in standard thin-walled vanadium sample containers, with temperature control being achieved with a standard helium closed-cycle cryostat.

Diffraction patterns were measured on the fixed-wavelength high-resolution powder diffractometer (HRPD) (Howard, Ball, Davis & Elcombe, 1983) at the HIFAR reactor of the Australian Nuclear Science and Technology Organisation, Lucas Heights, Australia. The patterns were recorded using neutrons of wavelength 1.884 Å.\*

### 3. Data analysis

Crystallographic data are collected in Table 1. Rietveld refinements were achieved using the LHPM3 program (Hill & Howard, 1986). The unit-

Table 1. *Crystallographic data and refinement details with e.s.d.'s in parentheses*

Compound	$\text{Ni}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2(\text{C}_6\text{D}_6)$ at 300 K	$\text{Ni}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2(\text{C}_6\text{D}_6)$ at 25 K	$\text{Zn}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2(\text{C}_6\text{H}_6)$ at 25 K
Space group	$P4/m$ (No. 83)	$P4/m$ (No. 83)	$P4/m$ (No. 83)
Lattice parameters:			
$a = b$ (Å)	7.2358 (6)	7.2196 (3)	7.3294 (3)
$c$ (Å)	8.3104 (8)	8.1007 (5)	8.0722 (4)
Neutron wavelength (Å)	1.884	1.884	1.884
$2\theta$ range (°)	16–150	16–150	16–150
$2\theta$ scan step (°)	0.06	0.05	0.05
No. of reflections	275	267	272
$R_{\text{wp}}$ (%)	4.48	4.83	3.78
$R_{\text{exp}}$ (%)	3.84	3.49	3.40
$R_B$ (%)	2.0	2.19	1.66

Table 2. *Atomic positions and thermal parameters (Å<sup>2</sup>) with e.s.d.'s in parentheses*

	$x$	$y$	$z$	$U$
$\text{Ni}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2(\text{C}_6\text{D}_6)$ at 300 K				
Ni(1)	0	0	0	0.0217 (14)
Ni(2)	$\frac{1}{2}$	$\frac{1}{2}$	0	0.0217 (14)
N(1)	0.2080 (12)	0.2032 (13)	0	0.0217 (14)
N(2)	0	0	0.2493 (17)	0.0217 (14)
C(1)	0.3176 (20)	0.3219 (18)	0	0.0217 (14)
C(2)	0	$\frac{1}{2}$	0.3400 (18)	0.0608 (37)
C(3)	0.1431 (16)	0.4302 (17)	0.4200 (12)	0.0748 (49)
D(2)	0	$\frac{1}{2}$	0.2094 (16)	0.0619 (38)
D(3)	0.2598 (22)	0.3733 (18)	0.3547 (18)	0.0981 (53)
H(4)	0.1164 (37)	0.0672 (21)	0.3001 (36)	0.1017 (142)
H(5)	0.0672 (21)	0.1164 (37)	0.3001 (36)	0.1017 (142)
H(6)	0	0.1344 (43)	0.3001 (36)	0.1017 (142)
$\text{Ni}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2(\text{C}_6\text{D}_6)$ at 25 K				
Ni(1)	0	0	0	0.0032 (9)
Ni(2)	$\frac{1}{2}$	$\frac{1}{2}$	0	0.0032 (9)
N(1)	0.2063 (8)	0.2037 (8)	0	0.0032 (9)
N(2)	0	0	0.2598 (11)	0.0032 (9)
C(1)	0.3183 (12)	0.3236 (12)	0	0.0032 (9)
C(2)	0	$\frac{1}{2}$	0.3296 (10)	0.0125 (18)
C(3)	0.1497 (9)	0.4293 (10)	0.4148 (8)	0.0272 (22)
D(2)	0	$\frac{1}{2}$	0.1963 (9)	0.0214 (20)
D(3)	0.2669 (12)	0.3741 (10)	0.3482 (11)	0.0303 (20)
H(4)	0.1176 (21)	0.0679 (12)	0.3166 (23)	0.0324 (66)
H(5)	0.0679 (12)	0.1176 (21)	0.3166 (23)	0.0324 (66)
H(6)	0	0.1357 (24)	0.3166 (23)	0.0324 (66)
$\text{Zn}(\text{NH}_3)_2\text{Ni}(\text{CN})_4 \cdot 2(\text{C}_6\text{H}_6)$ at 25 K				
Zn	0	0	0	0.0032 (12)
Ni	$\frac{1}{2}$	$\frac{1}{2}$	0	0.0032 (12)
N(1)	0.2088 (10)	0.2076 (10)	0	0.0032 (12)
N(2)	0	0	0.2659 (10)	0.0032 (12)
C(1)	0.3220 (14)	0.3237 (14)	0	0.0032 (12)
C(2)	0	$\frac{1}{2}$	0.3290 (10)	0.0063 (19)
C(3)	0.1475 (10)	0.4304 (10)	0.4145 (7)	0.0081 (16)
H(2)	0	$\frac{1}{2}$	0.1946 (21)	0.0322 (42)
H(3)	0.2634 (23)	0.3758 (18)	0.3473 (19)	0.0272 (39)
H(4)	0.1164 (28)	0.0672 (16)	0.3119 (26)	0.0377 (86)
H(5)	0.0672 (16)	0.1164 (23)	0.3119 (26)	0.0377 (86)
H(6)	0	0.1343 (33)	0.3119 (26)	0.0377 (86)

cell, zero-point, overall scale factor, peak-width/asymmetry and preferred orientation parameters were refined simultaneously with the structural and thermal parameters. A slightly preferred orientation was found along the [011] direction. The observed

\* A list of neutron powder diffraction data has been deposited with the IUCr (Reference AL0564). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

and calculated diffraction patterns are illustrated for Ni-Ni.C<sub>6</sub>D<sub>6</sub> (25 and 300 K) and Zn-Ni.C<sub>6</sub>H<sub>6</sub> (25 K) in Figs. 1(a-c), respectively, and the measures-of-fit are collected in Table 1. Atomic positions and isotropic thermal parameters are given in Table 2.

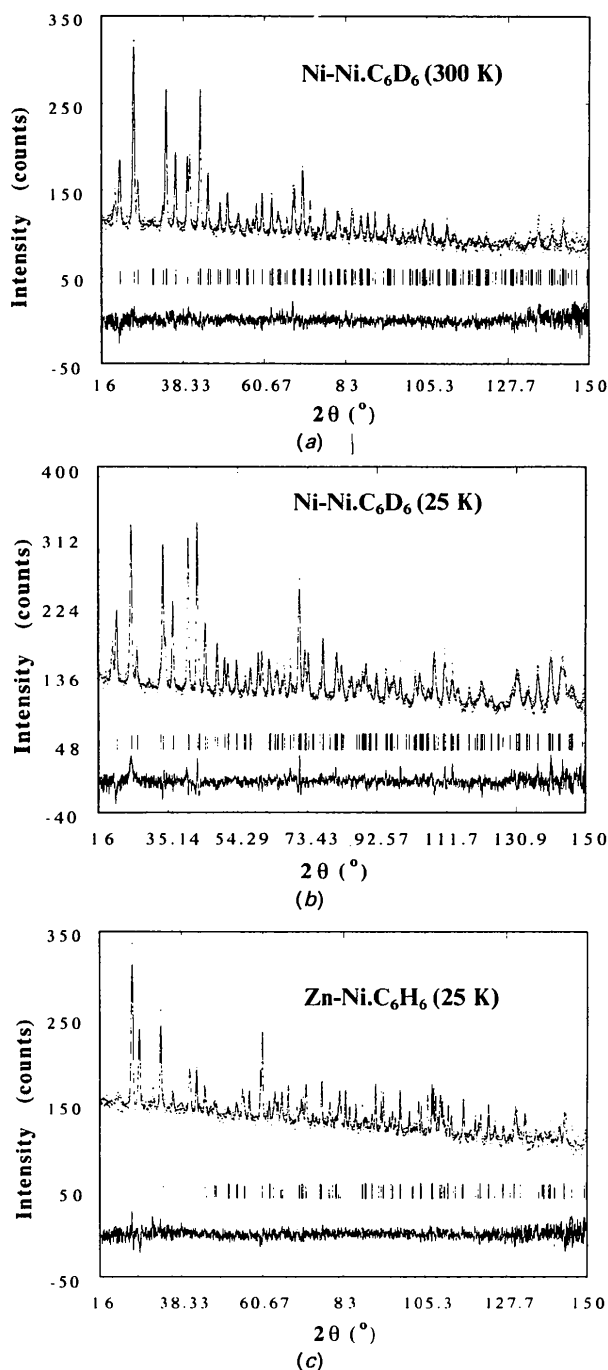


Fig. 1. Observed (crosses), calculated (line) and the difference (lower line) diffraction patterns for Ni(NH<sub>3</sub>)<sub>2</sub>Ni(CN)<sub>4</sub>.2C<sub>6</sub>D<sub>6</sub> at (a) 25, (b) 300 and (c) Zn(NH<sub>3</sub>)<sub>2</sub>Ni(CN)<sub>4</sub>.2C<sub>6</sub>H<sub>6</sub> at 25 K. The vertical lines mark the positions of the Bragg reflections.

## 4. Discussion

### 4.1. Description of structure

The structure of Ni-Ni.C<sub>6</sub>D<sub>6</sub> and Zn-Ni.C<sub>6</sub>H<sub>6</sub> was found to be consistent with that obtained from earlier X-ray diffraction studies (Rayner & Powell, 1952; Sasaki, 1969). The space group is *P4/m*, with one formula unit in the unit cell. The metal atoms form an infinite-plane network with the cyanide groups forming bridges *via* the N atom to the *M* atom and *via* the C atom to the *M'* atom. An octahedral configuration of the *M* atom is completed by the two axial NH<sub>3</sub> groups. The planes of the benzene rings are perpendicular to the plane of the metal atoms, as shown in Fig. 2.

### 4.2. Orientation of benzene molecules

In the refinements the benzene molecules were constrained to the molecular dimensions reported for solid benzene (David, Ibberson, Jeffrey & Ruble, 1992), with the angle formed between the molecular plane and the *a* axis being a refinable parameter. The refined value of this angle is 64°45' for both Ni-Ni.C<sub>6</sub>D<sub>6</sub> and Zn-Ni.C<sub>6</sub>H<sub>6</sub> at 25 K, which is to be compared with the value of 58°30' reported in a previous study of the Cd-Ni.C<sub>6</sub>H<sub>6</sub> analogue.

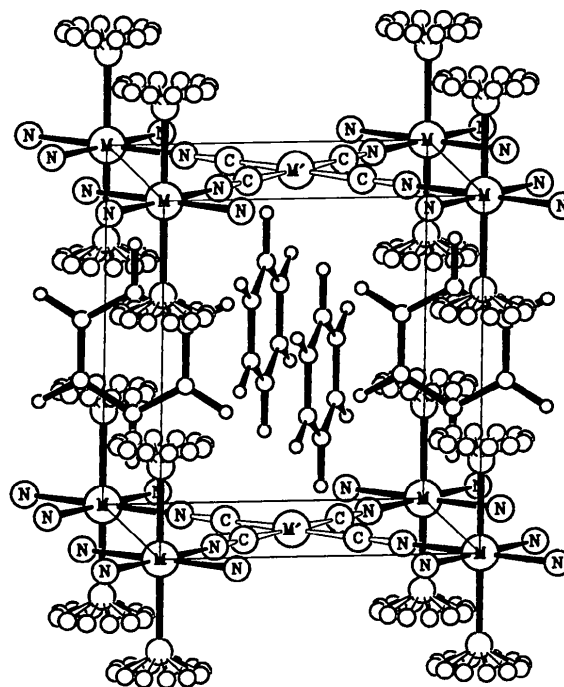


Fig. 2. Illustration of the crystal structure of *M*(NH<sub>3</sub>)<sub>2</sub>-*M'*(CN)<sub>4</sub>.2C<sub>6</sub>H<sub>6</sub>, showing the disorder of the NH<sub>3</sub> groups and the orientation of the benzene molecules. The thin line denotes the unit cell.

Table 3. Bond lengths (Å) and interbond angles (°) with e.s.d.'s in parentheses

	Ni(NH <sub>3</sub> ) <sub>2</sub> Ni(CN) <sub>4</sub> - 2(C <sub>6</sub> D <sub>6</sub> ) at 300 K	Ni(NH <sub>3</sub> ) <sub>2</sub> Ni(CN) <sub>4</sub> - 2(C <sub>6</sub> D <sub>6</sub> ) at 25 K	Zn(NH <sub>3</sub> ) <sub>2</sub> Ni(CN) <sub>4</sub> - 2(C <sub>6</sub> H <sub>6</sub> ) at 25 K
Ni(1)Zn—N(1)	2.104 (9)	2.093 (6)	2.158 (7)
Ni(1)Zn—N(2)	2.072 (14)	2.105 (9)	2.146 (8)
N(1)—C(1)	1.169 (16)	1.184 (11)	1.188 (13)
C(1)—Ni(2)/Ni	1.844 (14)	1.828 (9)	1.836 (10)
N(2)—H(4)	1.060 (26)	1.082 (15)	1.052 (19)
C(2)—C(3)	1.330 (13)	1.380 (8)	1.380 (8)
C(2)—D(2)/H(2)	1.085 (20)	1.080 (10)	1.085 (19)
C(2)—C(3)—D(3)/H(3)	120	120	120
Ni(1)Zn—N(1)—C(1)	177.05 (0.96)	177.67 (0.61)	179.15 (0.74)
Ni(1)Zn—N(2)—H(4)	113.46 (1.7)	115.11 (1.05)	110.67 (1.2)
Ni(2)Ni—C(1)—N(1)	177.03 (1.19)	177.20 (0.76)	179.04 (0.88)

#### 4.3. Structural changes between 25 and 300 K

The pattern of Ni-Ni.C<sub>6</sub>D<sub>6</sub> at 300 K could not be fitted satisfactorily with the constrained benzene molecule described above. However, when the constraints on bond lengths were removed, whilst retaining the constraint for a planar hexagonal unit, a satisfactory refinement was achieved. The main difference between the benzene molecules in Ni-Ni.C<sub>6</sub>D<sub>6</sub> at 25 and 300 K is a foreshortening of the C—C distance, due to thermal motion, from an apparent 1.38 to 1.33 Å, respectively. Bond lengths and angles are given in Table 3.

The length of the *a* axis increased by less than 0.02 Å (0.25%) on warming the sample from 25 to 300 K, whilst the *c* axis increases by more than 0.2 Å (2.5%); this expansion being mainly associated with an increased separation of the NH<sub>3</sub>NiNH<sub>3</sub> units. The lack of change in the *a* axis is consistent with the negligible change of angle between the plane of the benzene molecules and this axis over the same temperature range. The slightly longer *a* axis of the Zn analogue is expected as a result of the substitution of a slightly larger metal atom in the metal-atom planes.

#### 4.4. NH<sub>3</sub> free rotation

There are four crystallographically equivalent H-atom positions around each ammine N atom. Since the NH<sub>3</sub> groups are known to undergo free rotation even at 25 K, we use the scattering density of an NH<sub>3</sub> group which is rotationally disordered over four equivalent orientations. These orientations were constrained to be related by a rotation of  $\pi/6$ , as illustrated in Fig. 2, with the radius of rotation and the distance from the N atom to the centre of the H<sub>3</sub> triangle being allowed to refine. Because of the rather approximate treatment of the H-atom thermal parameters, there are large uncertainties in the positional parameters of the ammine H atoms, but nevertheless, the refined radius of rotation, 0.980 Å, compares well with the value of 0.985 Å derived from the rotational spectrum.

#### 4.5. Thermal parameters

Because of the presence of H atoms in the NH<sub>3</sub> groups of the Ni-Ni.C<sub>6</sub>D<sub>6</sub> sample and in the C<sub>6</sub>H<sub>6</sub> group of the Zn-Ni.C<sub>6</sub>H<sub>6</sub> sample, the diffraction patterns suffer from substantial background contributions. Consequently, we restricted the number of variables by using a single isotropic temperature factor for both metal atoms, both N atoms and the C atom of the cyanide group. An improved fit was obtained when the atoms of the benzene molecule were allowed individual temperature factors and this resulted in significantly higher temperature factors for the C(3) and H(3) [D(3)] atoms compared with the C(2) and H(2) [D(2)] atoms. This probably reflects some low-frequency whole-body libration of the benzene molecule, even at 25 K.

The use of split-atom positions and isotropic temperature factors is somewhat artificial for the NH<sub>3</sub> free rotors, but nevertheless, the rather large thermal parameters which result from the refinement confirm the presence of free rotation. Further, it was found that when the split-atom model is replaced by a single H atom [H(4)], this scattering centre can be placed anywhere on the circle described by the NH<sub>3</sub> rotations without any effect on the *r*-factors.

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## Synchrotron X-ray Study of the Electron Density in $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>

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### Abstract

Structure factors for synthetic haematite,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, have been measured for two small crystals using focused  $\lambda = 0.7$  Å synchrotron radiation. The structure factors from the two data sets are consistent. Approximate symmetry in the concordant densities, related more closely to the Fe—Fe geometry than to the nearest-neighbour Fe—O interactions, is similar to that in the corundum  $\alpha$ -Al<sub>2</sub>O<sub>3</sub> structure. Deformation density maxima are located at the midpoint of the Fe—Fe vector along the *c* axis, on a common face for O-octahedra, perpendicular to *c*. Maxima also occur at the midpoint of the Fe—Fe vector bisecting the edges of the O-octahedra. These results are in accordance with theoretical predictions for metal–metal bonding. Space group  $R\bar{3}c$ , hexagonal,  $M_r = 159.7$ ,  $a = 5.0355$  (5),  $c = 13.7471$  (7) Å,  $V = 301.88$  (7) Å<sup>3</sup>,  $Z = 6$ ,  $D_x = 5.270$  Mg m<sup>-3</sup>,  $\mu_{0.7} = 13.68$  mm<sup>-1</sup>,  $F(000) = 456$ ,  $T = 293$  K,  $R = 0.019$ ,  $wR = 0.021$ ,  $S = 5.55$  for 368 unique reflections in the most accurate data set 2.

### Introduction

The deformation density ( $\Delta\rho$ ) near the Al cations for  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, observed by Maslen, Streltsov, Streltsova, Ishizawa & Satow (1993), has approximate sixfold symmetry. That unexpected topography may be significant to our basic understanding of bonding in solids. It should, therefore, be tested by imaging the electron densities in related compounds. The *c/a* ratios for most corundum oxide structures, including

$\alpha$ -Al<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, Cr<sub>2</sub>O<sub>3</sub>, Ga<sub>2</sub>O<sub>3</sub> and Rh<sub>2</sub>O<sub>3</sub>, range from 2.70 to 2.74 (Prewitt, Shannon, Rogers & Sleight, 1969). That ratio is anomalously small, 2.6388 (3), for Ti<sub>2</sub>O<sub>3</sub> (Vincent, Yvon, Grüttner & Ashkenazi, 1980) and anomalously large, 2.8280 (3), for V<sub>2</sub>O<sub>3</sub> (Vincent, Yvon & Ashkenazi, 1980). The corundum-type  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> unit cell is bigger than that for  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>, but the axial ratio *c/a*, determined to be 2.731 (1) for crystal (1) and 2.7300 (4) for crystal (2), is within 1 e.s.d. of *c/a* = 2.7308 (3) for  $\alpha$ -Al<sub>2</sub>O<sub>3</sub>.

According to a schematic band structure proposed by Goodenough (1971) for corundum-type *3d* crystals, the trigonally distorted octahedral crystal field quantized along the threefold axis splits the fivefold-degenerate *3d* state into a stable  $\sigma$ -bonding pair of  $e_g$  orbitals directed toward the nearest-neighbour O atoms, a pair of  $e_t$  orbitals directed approximately towards three neighbouring cations in a puckered basal plane through common octahedral site edges, and an  $a_t$  orbital directed along the *c* axis towards one near-neighbour through a common octahedral site face. From symmetry considerations alone, the energy band scheme shows that the bonding  $a_t$  and  $e_t$  band overlap depends on the *c/a* ratio and on the widths of these bands.

In theory, the  $a_t$  and  $e_t$  orbitals for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are both half filled. Any variation in the *c/a* ratio that stabilized one would be at the expense of the other. The small *c/a* ratio of Ti<sub>2</sub>O<sub>3</sub> stabilizes a filled and bonding  $a_t$  band at the expense of empty  $e_t$  bands. The large *c/a* ratio of V<sub>2</sub>O<sub>3</sub> stabilizes the two bonding  $e_t$  orbitals per cation relative to the lower  $a_t$  band. Experimentally determined  $\Delta\rho$  maps for Ti<sub>2</sub>O<sub>3</sub> (Vincent, Yvon, Grüttner & Ashkenazi, 1980) and for V<sub>2</sub>O<sub>3</sub> (Vincent, Yvon & Ashkenazi, 1980) support

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